

# In situ atomic-scale observation of dislocation-mediated discrete plasticity in nanoscale crystals

## Editorial

Nanoscale metals with superior mechanical properties are promising building blocks in the next generation flexible electronics.<sup>1</sup> Due to the pronounced surface effect caused by the ultrahigh surface-to-volume ratio of nanocrystals, the mechanical behaviors of nanoscale metals are quite different from their bulk counterparts. Dislocation nucleation is demonstrated to be related to the microstructural length scale of materials. With decreasing the feature size, dislocation source changed from Frank-Read source to single-arm dislocation source.<sup>2</sup> As sample size decrease to nanoscale, surface dislocation

nucleation is the dominant mechanism. In situ transmission electron microscope (TEM) nanomechanical testing opens up an avenue to directly observe the atomic-scale dislocation dynamics in nanoscale metals. Here we report an in situ atomic-scale observation of dislocation-mediated discrete plasticity in gold (Au) and tungsten (W) nanowires (NW).<sup>3,4</sup> In the tensile test of Au NW loaded along [001]

under a strain rate of  $10^{-3} \text{ s}^{-1}$  and viewed along [110], surface steps, consisted of two sets of {111} planes, can act as dislocation nucleation sites. As the stress inside the sub-10-nm Au NW is high enough to induce yielding, a leading partial dislocation nucleated from a surface step, and then glided through the Au NW, leaving behind a stacking fault (SF), as shown in Figure 1A-1B. Subsequently, this SF was eliminated by a trailing partial emitted from the same dislocation nucleation site as the previous leading partial (Figure 1C). Hence, discrete dislocation nucleation and propagation mediated plasticity in Au NW. In addition to Au NW, surface dislocation nucleation was also dominant in W NW under <112> compression viewed along [111].<sup>4,5</sup> After yielding, dislocations nucleated from free surface, then glided on the {110} slip planes, and finally annihilated at the opposite free surface (Figure 1D-1G). Dislocation nucleation behavior in W NW was strain-dependent. At a low compressive strain level,  $1/2\langle 111 \rangle$ -type mixed dislocations nucleated from (110) surface

Volume 5 Issue 3 - 2021

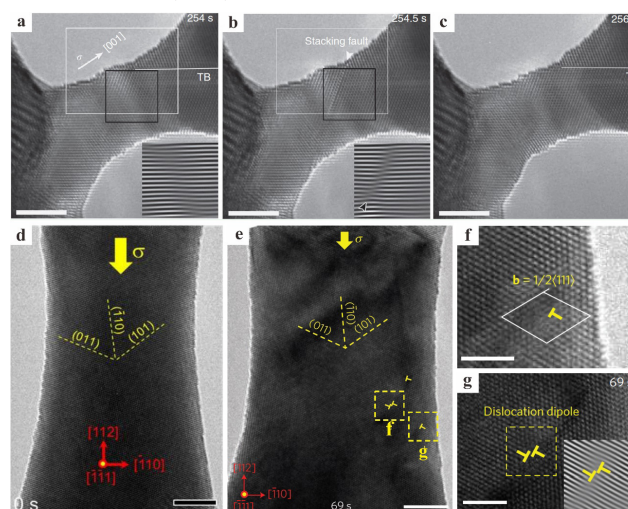
Sixue Zheng, Scott X Mao

Department of Mechanical Engineering and Materials Science, University of Pittsburgh, USA

**Correspondence:** Scott X Mao, Department of Mechanical Engineering and Materials Science, University of Pittsburgh, Pittsburgh, Pennsylvania 15261, USA, Email [sxm2@pitt.edu](mailto:sxm2@pitt.edu)

**Received:** April 30, 2021 | **Published:** May 12, 2021

(Figure 1F); however, at higher strain level, half dislocation loops, that is dislocation dipoles, heterogeneously nucleated from (111) surface (Figure 1G). Such strain-dependent nucleation behaviors can be attributed to the difference in surface morphologies for (110) and (111) planes. The (111) surface planes were zigzag, while the (110) surfaces were atomically flat. At high strain level, there exist local strain concentration at surface facets of (111) planes, favoring surface dislocation nucleation. Similar to face-centered cubic (FCC) and body-centered cubic (BCC) nanoscale metals, surface dislocation nucleation is also dominant in other nanostructured metals, including Au NWs with nanotwins, bi- and penta-twinned silver (Ag) NWs, and bicrystalline Ag NWs.<sup>6</sup> In summary, our work offers atomic-scale insights into surface dislocation nucleation and the related dislocation dynamics in nanoscale metals, providing guidelines to fabricate nanomaterials with high performance to achieve their potential applications in nanoscale devices. To date, the effect of chemical passivation layer on surface dislocation nucleation remains experimentally unclear, which probably attracts a great deal of interest for future research.



**Figure 1** Dislocation-mediated discrete plasticity in nanoscale metals. (A-C) Sequential TEM images showing dislocation dynamics. The insets in a and b are Fourier-filtered images of the black square area. Scale bars in a-c are 3nm. (D-E) TEM images of dislocation dynamics in W NW. (F-G) TEM images of  $1/2\langle 111 \rangle$ -typed mixed dislocation loop nucleated from free surface. Scale bars in d-e and f-g are 5nm and 2nm, respectively. Source: (A-C) reprinted with the permission from Ref.<sup>3</sup> © 2010, Nature Publishing Group. (D-G) reprinted with the permission from Ref.<sup>4</sup> © 2015, Nature Publishing Group.

## Acknowledgments

S.X.M. acknowledges support from the NSF 1760916 through University of Pittsburgh.

## Conflicts of interest

The authors declare that there is no conflict of interest.

## References

1. Zheng SX, Luo XM, Wang D, et al. A novel evaluation strategy for fatigue reliability of flexible nanoscale films. *Mater Res Express*. 2018;5(3):035012.
2. Oh SH, Legros M, Kiener D, et al. In situ observation of dislocation nucleation and escape in a submicrometre aluminium single crystal. *Nature materials*. 2009;8(2):95.
3. Zheng H, Cao A, Weinberger CR, et al. Discrete plasticity in sub-10-nm-sized gold crystals. *Nature Communications*. 2010(1).
4. Wang J, Zeng Z, Weinberger CR, et al. In situ atomic-scale observation of twinning-dominated deformation in nanoscale body-centred cubic tungsten. *Nature materials*. 2015;14(6):594–600.
5. Wang J, Wang Y, Cai W, et al. Discrete shear band plasticity through dislocation activities in body-centered cubic tungsten nanowires. *Scientific Reports*. 2018(8):4574.
6. Zheng S, Mao SX. Advances in experimental mechanics at atomic scale. *Extreme Mechanics Letters*. 2021:101284.