

Research Article





Study of electronic and lattice dynamical properties of antiperovskite-type nitrides XNNi₃ (X= Pd, Sn and Sb)

Abstract

First principles study of electronic and lattice dynamical properties of the XNNi. (X=Pd. Sn and Sb) ternary nitrides with E2₁ structure (space group Pm3 m (221)) has been reported using the plane-wave pseudo-potential technique based on density functional theory. The calculated equilibrium parameters are in good agreement with other works. The relationship between anisotropy and mechanical properties are also analyzed. Mechanical stability and stiffness of these materials are determined and XNNi₃ (X=Pd, Sn and Sb) ternary nitride compounds are found mechanically stable at zero pressure. Shear Modulus (G), Young's Modulus (E), maximum and minimum Poisson ratios (v), Zener anisotropy factor (A) and compressibility (β) values are calculated and evaluated in calculations of elastic properties. The electronic properties are studied and presented by plots with total and partial density of states with charge density distributions. The XNNi, (X=Pd, Sn and Sb) ternary nitrides are metallic behavior and have covalent bonding due to the hybridization. The vibrational properties are investigated to explain lattice dynamics of these types of ternary nitrides.

Keywords: first-principles, lattice dynamical properties, electronic properties, ternary nitrides, antiperovskite

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Introduction

The antiperovskite type Ni-rich ternary nitrides XNNi, researches have been increased since the discovery of superconductivity (8 Kelvin) for cubic antiperovskite MgCNi, compound. This discovery has strongly motivated to study antiperovskite series. The studies of the family of Ni-rich carbides have been investigated theoretically and experimentally.2-15 In recent years, the investigations on some new antiperovskite type nitrides have gained wide interest also some biosensor applications in bioelectronics industry. 16-30 Despite of the fact that there are number of studies related with their properties for some cubic antiperovskite type Ni-rich ternary nitrides, in particular, the mechanical properties of XNNi,-type compounds with X= Al, Ga, In, Zn, Cd, Mg, Sn, Sb, Pd, Cu, Ag and Pt have been studied in theory, 15 it is not mentioned on especially lattice dynamical and electronic properties of XNNi₃ (X= Pd, Sn and Sb). One of interesting study is about a new Ni-based antiperovskite nitride, CuNNi, that shows the superconductivity at 3.2 K and it is reported with X-ray diffraction, magnetization, resistivity and heat capacity measurements.¹⁶ The structural and mechanical properties of the antiperovskite XNNi, (X=Zn, Mg, Al) with pressure effect are studied by Hong-Cun et al., 26 by using CASTEP code. Optical functions of SnNNi, ZnNNi, and CuNNi, compounds are studied until 40 eV.29 In this study, by means of the ab-initio calculations, we have analyzed in details the comparative study of electronic and elastic properties of the XNNi, (X=Pd, Sn and Sb). Optimized lattice parameters and electronic band structures are reported by using ultra soft pseudo potential.31,32 In addition, anisotropic independent second order elastic constants (C_{ii}). These constants give permissions us to get the mechanical parameters of XNNi₃ (X=Pd, Sn and Sb). Additionally, vibrational properties of XNNi, (X=Pd, Sn and Sb) compounds are investigated and summarized.

Method of calculation

The density functional theory (DFT) 33,34 has successfully been applied to the ab-initio calculations of the ground-state properties. In view of these circumstances, we have applied to the Generalized Gradient Approximation (GGA)35 for the exchange-correlation functional. All properties of calculations are investigated by using the Vienna Ab-initio Simulation Package (VASP). 36-39 The calculations are performed for $Pd(4d^{10})$, $Sn(5s^25p^2)$, $Sb(5s^25p^3)$, $N(2s^22p^3)$, Ni(4s¹3d⁹). In our calculations, plane-wave basis sets with cut-off energy 500 eV and the 12x12x12 Monkhorst and Pack⁴⁰ k-points are used in the Brillouin zone for XNNi, (X=Pd, Sn and Sb). To obtain mechanical anisotropic properties, EIAM code is used for calculations. 41 The elastic properties are exploited to estimate with stress-strain method.42,43

Results and discussion

Structural and elastic properties

The unit cell of SnNNi, compound is shown in Figure 1. The crystal structures of SbNNi, and PdNNi, compounds are the same with SnNNi, compound as shown in Figure 1. In our case E2,-type structure which is illustrated in Figure 1. The Wyckoff positions of atoms are located as follow: Sn (0, 0, 0); N (0.5, 0.5, 0.5); and Ni (0, 0.5, 0.5) (Figure 2). Firstly, the equilibrium lattice constants, bulk modulus and its pressure derivative have been obtained by minimizing the total crystal energy calculated for different values of lattice constants using the Birch-Murnaghan equation of states (eos)44 and the calculation results are given in Table 1 for cubic perovskite (E2,) structure (space group Pm3 m (221)) of XNNi₃ (X=Pd, Sn and Sb). (Table 1) the present structural results are listed in the Table 1, along with the other theoretical and experimental works. The present lattice constants are obtained as 3.905 Å, 3.944 Å and 3.808 Å, respectively,



for XNNi₃ (X=Pd, Sn and Sb) compounds. Our lattice constants are very good convenient parameters with the other theoretical studies. Our lattice constants in E2,-type crystal structure for SnNNi, is nearly 0.127% lower, for SbNNi, is nearly 0.051% higher and for PdNNi, is nearly 0.132% higher than the reference value.²⁸ These deviations may stem from the using of GGA approximations with different abinitio codes. Additionally, the volume values and bulk modulus values of SnNNi, compound are in convenient with other theoretical values and also the same values are in very good agreement of SbNNi, and PdNNi, compounds. The effect of hydrostatic pressure indicated by the derivative of bulk modulus under pressure (B') is given in Table 1 for each of ternary nitride compounds. According to Table 1, the pressure derivatives of bulk modulus are calculated 4.440, 4.399 and 4.590, respectively, for XNNi, (X=Pd, Sn and Sb) compounds and sequenced like as $B'(SbNNi_3) > B'(PdNNi_3) > B'(SnNNi_3)$. The derivative of bulk modulus is evaluated with anisotropy. From Table 3, the anisotropy values of XNNi₃ (X=Pd, Sn and Sb) compounds are, respectively, 1.19, 1.28 and 1.21. The magnitudes of anisotropy factors are arranged with $A(SbNNi_3) > A(PdNNi_3) > A(SnNNi_3)$. Due to both of these equalities, derivative of bulk modulus are confirmed by anisotropy as an expected. The thermodynamic stability of XNNi, (X=Pd, Sn and Sb) compounds can be reflected by the formation enthalpy (AH). Negative formation enthalpy has been explained as an exothermic process, and formation energy in the lower ones shows the stability related with the decomposition to the constituents of an element. The formation enthalpy could be expressed by the relation:⁴⁵

$$\Delta H = (E_{top} - (\Sigma n_i E_i) / n$$
 (1)

where E_{tot} is the total energy of the compound with n_i atoms of all i (X(Pd, Sn, Sb), N and Ni). n: total number of atoms in the primitive cell, E: total energy of a pure i the atom with equilibrium lattice constants. The calculated theoretical formation enthalpies of XNNi, (X=Pd, Sn and Sb) compounds are included in Table 1. As far as we known, there are no data for evaluation the formation energy in the literature to compare with ours. SbNNi, shows the lowest value of formation enthalpy, which indicates that SbNNi₃ compound has the highest stability of these nitride structures. It is important to investigate the second order elastic properties because of the fact that the calculations provide an accuracy and comparison of the calculations of mechanical properties. Herein, C;; elastic constants are the second-order elastic constants of the structure and has been optimized under a given set of exchange-correlation (XC) potential functions and attained an equilibrium structure with a minimum total energy. The elastic parameters are obtained from the second-order derivatives of the total energy:

$$C_{ij} = \frac{1}{V_0} \frac{\partial^2 E_{total}}{\partial \xi_i \partial \xi_j} \tag{2}$$

The cubic crystal has only three independent parameters, C_{II} , C_{I2} and C_{44} . The traditional rules on the elastic constants: $C_{II} > 0$, $C_{II} - C_{I2} > 0$, $C_{44} > 0$, $C_{II} + 2C_{12} > 0$ and $C_{II} > B > C_{12}$. These traditional mechanical stability conditions (called that Born's stability criteria) $(P=0\ GPa)^{46}$ are investigated by using the obtained second-order elastic constants all our three nitride compounds. The calculated values of C_{ij} are summarized and given in the table for XNNi₃ (X=Pd, Sn and Sb), respectively (Table 2). Second-order elastic constants of XNNi₃ (X=Pd, Sn and Sb) meant to Born's stability conditions Table 2. According to Table 2, it is obvious that XNNi₃ (X=Pd, Sn and Sb)

compounds satisfy stability conditions. For SbNNi₃ compound in reference, 28 C₄₄ is found as -8.6 GPa although our present calculated value is 34.79 GPa for C₄₄. All other theoretical references are compatible with present values. As it can be seen from Table 2, our ternary nitride compounds have different elastic constants due to their classifications of elements. Elastic properties of our nitride compounds are effected owing to the fact that Tin (Sn) is post-transition metal, antimony (Sb) is metalloid and palladium (Pd) is transition metal. The Zener anisotropy factor (A), Poisson ratio (δ), and Young's modulus (E) that are important parameters to see all image of elastic properties are also calculated using by these formulas: 47

$$A = \frac{2C_{44}}{C_{11} - C_{12}} \tag{3}$$

$$\tilde{o} = \frac{1}{2} \left[\frac{(B - \frac{2}{3}G)}{(B + \frac{1}{3}G)} \right], \tag{4}$$

$$E = \frac{9GB}{G + 3B} \tag{5}$$

where G is the an isotropic shear modulus as a function of crystal orientation and is given like that $G=(G_V+G_R)/2$, herein G_V is Voigt's shear modulus (it is related with the upper bound of G values) and G_R is Reuss's shear modulus (it is related with the lower bound of G values) and can be written as $G_V = (C_{II} - C_{I2} + 3C_{44})/5$ and $5/G_R = 4/(C_{II} - C_{I2}) + 3/6$ C_{44} , resopectively. The calculated an isotropic shear modulus, Young's modulus, Poisson ratios, Zener anisotropy factor, and compressibility (β) of the XNNi, (X=Pd, Sn and Sb) are presented in Table 3. The Shear and Young's modulus are calculated with their Voigt and Reuss values and Poisson ratios with maximum and minimum values at zero pressure by ElAM code⁴¹ for anisotropic behaviors of three ternary nitrides (Table 3). Obtained by Voigt and Reuss values of isotropic shear modulus (G) are 152.21 GPa, 129.39 GPa and 129.11 GPa, respectively, for PdNNi₂, SnNNi₃ and SbNNi₃ compounds. Using ratio of isotropic shear modulus and bulk modulus, elastic manners of materials are estimated. Ratios of G/B that is called Pugh ratio of XNNi, (X=Pd, Sn and Sb) compounds are given in Table 3. Providing that G/B<0.5, the material exhibits in a ductile behavior, and while G/ B>0.5, the material exhibits in a brittle behavior. 47,48 As can be seen in Table 3, all of XNNi₃ (X=Sn, Sb and Pd) ternary nitrides compounds indicate brittle manners due to the fact that their G/B ratios are greater than 0.5. In fact, they behave nearly at brittle/ductile border like in reference.¹³ It has also observed that for all of our three antiperovskite type nitrides B>G. As mentioned that parameters limit the mechanical stability of these materials. As a comparison, the Young's modulus of PdNNi, compound has the biggest one in our ternary nitrides systems. From the literature it is well-known that, if the rate of Poisson is less than 0.25, the material shows covalent bond character, otherwise it is bigger than or equal to 0.25 it shows ionic bond character.⁴⁹ The minimum value Poisson's ratio of SbNNi, is calculated as zero and maximum value Poisson's ratio of SbNNi, is calculated as 0.14. It might have originated from directions or maximum stability. The other values of Poisson's ratios of XNNi, (X=Pd, Sn and Sb) are obtained similar values for each of three nitride compounds. Three ternary E2, structure nitrides show metallic-like systems as indicated in reference.¹³ In bulk materials, to see the elastic anisotropy behavior, the Zener anisotropy factor is using to determine the degree of anisotropy. Providing that it gives the value of 1, our compound shows entirely isotropic. Otherwise, this value exhibits anisotropic behavior. The values of our three nitrides are greater than 1. Our materials partially exhibit anisotropic behaviors. The compressibility is a measure of elasticity and is defined as following relations:⁵⁰

$$\beta = \frac{C_{11} - C_{12}}{\Omega} \tag{6}$$

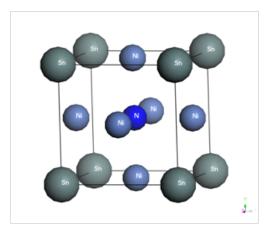
$$\Omega = (C_{11} + C_{12})C_{11} - 2C_{12}^{2}$$
 (7)

The calculated compressibility values are found as $0.0023~GPa^{\text{-}1}$, $0.0029~GPa^{\text{-}1}$ and $0.0031~GPa^{\text{-}1}$, respectively, for PdNNi $_3$, SnNNi $_3$ and SbNNi $_3$ compounds. The calculated present values of compressibility are compatible with other theoretical data for SbNNi $_3$ and PdNNi $_3$ compounds.

Table I Calculated lattice parameter (a_{col}), volume (V), bulk modulus (B), pressure derivative of bulk modulus (dB/dP), formation energy (ΔH)

Material	a _{cal} (Å)	V (ų)	B (GPa)	dB/dP	∆ H (eV)	
PdNNi,	3.808	55.219	190.4	4.44	-2.977	
3	3.803⁵	55.002b	212.10 ^b			
	3.783°					
	3.809 ^d		217.40 ^d			
SnNNi,	3.905	59.547	159.203	4.399	-3.095	
3	3.768 ^a	53.497a				
	3.910⁵	59.776 ^b	182.30 ^b			
	3.927°	60.559°	176.80°			
C1	2011	41.240	154 405	4.50	2.100	
SbNNi ₃	3.944	61.349	156.405	4.59	-3.109	
	3.942⁵	61.256 ^b				
	3.766°					

Abbreviations: ^a, numerical study according to empirical model, ⁸; ^b, theoretical study with APW+lo (FLAPW) implemented in WIEN2k code, GGA-PBE, ²⁸; ^c, theoretical study with CASTEP code, GGA, ²⁹; ^d, theoretical study with CASTEP code, GGA-PBE³⁰



 $\textbf{Figure I} \ \, \textbf{The unitcell of SnNNi}_{3}.$

Table 2 Second-order elastic constants (C_{ii}), bulk modulus (B), stability

Material	C,, [GPa]	C ₁₂ [GPa]	C ₄₄ [GPa]	B [GPa]	Stability
PdNNi ₃	324.67 313.3° 315.6 ^b	164.5 161.5 ^a 168.3 ^b	53.71 81.1 ^a 42.8 ^b	190.4 212.1 ^a	Stable
SnNNi ₃	272.33 239.9 ^a	138.75 153.5 ^a	39.28 2.2 ^a	159.203 182.3a 176.8b	Stable
SbNNi ₃	257.48 226.7 ^a	142.52 154.3 ^a	34.79 -8.6 ^a	156.405	Stable

Abbreviations: a, GGA-PBE28; b, GGA29; c, GGA-PBE30

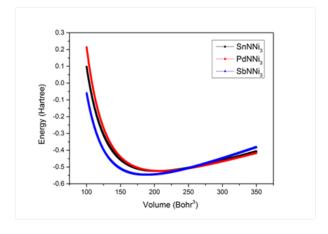


Figure 2 Total energy versus volume curves of XNNi₃ (X=Pd, Sn and Sb).

Table 3 Voigt's shear modulus (GV), Reuss's shear modulus (GR), voigt's young's modulus (EV), reuss's Young's modulus (ER), Maximum and minimum poisson ratios (ν), zener anisotropy factor (A), compressibility (β)

	PdNNi ₃	SnNNi ₃	SbNNi ₃
GV [GPa]	152.89	129.86	130.05
GR [GPa]	151.52	128.91	128.17
G/B	8.0	0.81	0.83
EV [GPa]	338.8	284.35	279.15
ER [GPa]	336.54	282.83	276.25
$v_{\rm max}$	0.16	0.14	0.14
$v_{\scriptscriptstyle min}$	0.05	0.04	0
A	1.21 1.06 ^a 0.58 ^b	1.19	1.28
β[GPa ⁻¹]	0.0023 0.0047 ^a 0.0046 ^b	0.0029 0.0055 ^a	0.0031

Electronic properties

In this section, the main features of electronic properties of XNNi, (X=Pd, Sn and Sb) compounds are described by analyzing the density of states as total and partial with their related charge densities in Figure 3 & Figure 4. The energy zero is chosen to be at the Fermi energy E_r. All of the three total densities of states have nearly similar features. For all compounds conduction band minimum values are upper than from Fermi energy level. As a comparison of DOS of SbNNi, with other compounds, its DOS is lower at Fermi energy level. The DOS of PdNNi, are above at Fermi energy level according to SbNNi, compound. These all compounds exhibit metallic character in consideration of rate for impletion at Fermi energy levels. The metallic behavior of XNNi₂ (X=Pd, Sn and Sb) compounds are mostly owing to the addition of Ni-d states at the Fermi level and a little addition of Pd-d states for PdNNi, compound. It is clearly seen that Sn-s state and N-p state contribute at Fermi level, and this emerges to a sp-hybridization between metal-s state and N-p states. As it is seen from partial density of states explanations, owing to the covalent bonding, there is hybridization and clarifies the charge densities in Figure 4. The lower valance band is because of the 2s-states electrons for XNNi₃ (X=Pd, Sn and Sb) compounds. The charge densities of XNNi₂ (X=Pd, Sn and Sb) compounds are depicted in Figure 4. The computed charge density distributions are evident that the covalent bonding that nature of our three ternary nitrides is obtained as covalent due to the sp-hybridization that is also confirmed by partial density of states plots. It is easy to observe that from Figures 3 and 4, the SbNNi, compound is much more covalent according to PdNNi, and SnNNi, compounds. The charges are more accumulated between atoms. Moreover, a high ratio of G/B is related with brittleness. Considering that SbNNi, compound has the greatest value of G/B, consolidates that SbNNi, compound has more covalent character than PdNNi, and SnNNi,.

Additionally, stability of XNNi₃ compounds is also confirmed by Band Filling Theory.
^{51,52} Considering the Band Filling Theory, the numbers of bonding states increase, the stability of material increases and anti-bonding states decrease the stability of compounds. If we called the ratio the width of the occupied states (W_{occ}) and the width of bonding states (W_b), we can explain the work about the material stability. If the ratio of W_{occ}/W_b is closer to 1.0, the stability increases. In this work, these quantities predict the structural stability, namely, the pseudo-gaps (W_p), gaps of occupation (W_{occ}), gaps of bonding (W_b) and the W_{occ}/W_b values are calculated for each compound and presented in Table 4 for XNNi₃ compounds. Also shown in Table 4, using this band theory formulation, the ratio of W_{occ}/W_b equals 0.987 and is closer to 1 for SbNNi₃ compound. It is obvious that SbNNi₃ compound is the most stable material. This result confirms that the previous presented partial density of states for our nitrides and charge density distributions for our nitrides.

Vibrational properties

The phonon dispersion curves of XNNi₃ (X=Pd, Sn and Sb) were obtained using by the PHONOPY code.⁵³ The partial atomic phonon density of states (DOS) for XNNi₃ ternary nitrides were calculated along the high symmetry directions using a 2x2x2 super cell and given in Figure 5A-5C. The 0.03 Å for displacement is adopted for each atom of the 2x2x2 supercell in to determine the forces of the atoms. The primitive cells of XNNi₃ contains 5 atoms with 15

phonon branches have 3 acoustic and 12 optical modes. For PdNNi₃ compound, a gap between acoustic and optic modes is found in the phonon dispersion curves owing to the bigger ratio of mass cation and anions. But, accordin to the SnNNi, and PdNNi, compounds there is not a gap between acoustic and optic modes. The lack of soft phonon imaginariy modes in the phonon spectra that supports the stable character as dynamically for the XNNi₃ (X=Pd, Sn and Sb) nitrides. In the literature, there is no study of the lattice dynamics of these compounds to compare with our data. It can be seen from Figure 5 that the low-lying optical phonon modes have interactions with phonon modes for SnNNi, and SbNNi, as the acoustic. For the phonon DOS of XNNi, ternary nitrides the acoustic modes are emerged by the vibrations of Sn, Sb and Pd atoms, while the optical modes are emerged by the vibrations of Ni atoms at low modes, with less addition from N and Ni atoms. At higher optical mode, the main contributions emerge from N atoms, with less contributions comes from Ni atoms. PdNNi, has upper phonon energies than SnNNi, and SbNNi, at Gamma point. The main distinction of the three nitrides is due to difference in the chemical bonding and masses for XNNi, (X=Pd, Sn and Sb) nitrides (Figure 5).

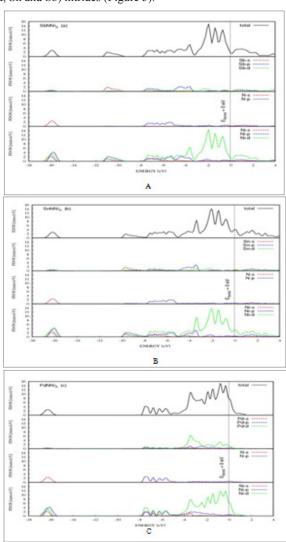


Figure 3 Partial and total DOS of A) PdNNi,, B) SnNNi, C) SbNNi,.

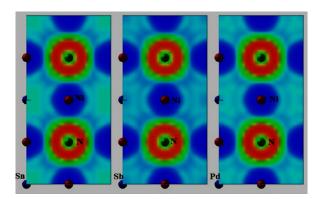
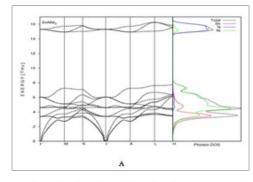
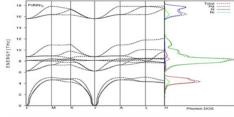


Figure 4 Charge density distribution of (110) plane of XNNi₃ (X=Pd, Sn and Sb) compounds.

Table 4 The calculated pseudogap W_p (eV), the width of occupied states W_{occ} (eV), bonding states W_b (eV), electron numbers at fermi levels n (Fermi) for XNNi, compounds

Materials	W _p	W _{occ}	$W_{_b}$	W_{occ}/W_{b}	n
PdNNi ₃	0.773	9.453	10.185	0.928	7.829
$SnNNi_3$	0.481	11.677	12.158	0.96	4.162
SbNNi ₃	0.166	13.082	13.248	0.987	1.526





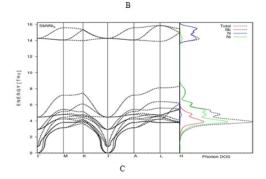


Figure 5 Calculated phonon dispersion curves and partial density of states throughout the high-symmetry directions in the BZ of XNNi3 (X=(A)Pd, (B) Sn and (C)Sb).

Conclusion

In this work, we have studies structural, elastic, electronic and vibrational properties of antiperovskite type nitrides XNNi, (X=Pd, Sn and Sb) compounds with E21 crystal structure using the GGA. The found lattice constants, volume, bulk modulus as structural parameters at zero pressure are in convenient with the previous work. Mechanical stability of XNNi, (X=Pd, Sn and Sb) compounds are predicted by Born's stability criteria and found that three nitrides show stability at zero pressure. In elastic calculations, isotropic shear modulus, Poisson's ratios and Young's modulus were estimated using with Voigt and Reuss approximations. In addition to good understand of mechanical behaviors of these compounds, anisotropy factor and compressibility are determined. From our first-principles calculations, the stoichiometric XNNi, (X=Pd, Sn and Sb) compounds are very similar in both structural and elastic properties. The three ternary nitrides have metallic behavior and exhibit covalent characters. The mechanical behavior of XNNi, (X=Pd, Sn and Sb) compounds are corroborated with electronic properties as given in the results section. The calculated phonon spectra and phonon DOS indicate that XNNi₃ (X=Pd, Sn and Sb) compounds are dynamically stable. To best of our knowledge is there is no experimental or theoretical study in vibrational properties of XNNi, (X=Pd, Sn and Sb) ternary nitrides has been reported yet for comparison. We predict that, our results are good and qualified estimations for future investigations.

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Conflicts of interest

There are no conflicts to declare.

References

- He T, Huang Q, Ramirez AP, et al. Superconductivity in the non-oxide perovskite MgCNi, Nature. 2001;411:54.
- 2. Park MS, Giim JS, Park Sh, et al. Supercond Sci Technol. 2004;17:274.
- Schaak RE, Avdeev M, Lee WL, et al. Formation of transition metal boride and carbide perovskites related to superconducting MgCNi₃. J Solid State Chem. 2004;177:1244.
- Dong AF, Che GC, Huang WW, et al. Synthesis and physical properties of AlCNi_x. *Physica C*. 2005;422:65.
- Tong P, Sun YP, Zhu XB, et al. Strong electron-electron correlation in the antiperovskite compound GaCNi₁. Phys Rev B. 2006;73:245106.
- Uehara M, Amano T, Takano S, et al. Chemical pressure effect on the superconductor MgCNi₃. *Physica C*. 2006;440:6.
- Uehara M, Yamazaki T, Kori T, et al. Superconducting Properties of CdCNi₃. J Phys Soc Jpn. 2007;76: 034714.
- Alexandrov KS, Beznosikov BV. Perovskites. Present and Future. Novosibirsk: 2004. p. 200–201.
- Johannes MD, Pickett WE. Electronic structure of ZnCNi₃. Phys Rev B. 2004;70:060507.
- Shein IR, Ivanovskii AL. Electronic and elastic properties of non-oxide anti-perovskites from first principles: Superconducting CdCNi₃ in comparison with magnetic InCNi₄. Phys Rev B. 2008;77:104101.
- Shein IR, Shein KI, Ivanovskii AL, et al. Metallofoz. Noveishie Tekhnol. 2004;26:1193.

- Shein IR, Bannikov VV, Ivanovskii AL. Structural, elastic and electronic properties of superconducting anti-perovskites MgCNi₃, ZnCNi₃ and CdCNi, from first principles. *Physica C*. 2008;468:1.
- 13. Bannikov VV, Shein IR, Ivanovskii AL. Phys Solid State. 2007;49:1704.
- 14. Dong AF, Che GC, Huang WW, et al. Synthesis and physical properties of AlCNi_x. *Physica C*. 2005;422:65.
- 15. Bannikov VV, Shein IR, Ivanovskii AL. J Struct Chem. 2010;51:170.
- 16. He B, Dong C, Yang L, et al. CuNNi₃: a new nitride superconductor withantiperovskite structure. *Supercond. Technol.* 2013;26:125015.
- Uehara M, Uehara A, Kozawa K, et al. New Antiperovskite-Type Superconductor ZnN, Ni., J Phys Soc Jpn. 2009;78:033702.
- Tutuncu H, Srivastava GP. Electronic structure, phonons and electron– phonon interaction in MgXNi₃ (X = B, C and N). J Phys.: Condens Matter. 2006:18:11089.
- Karki AB, Xiong YM, Young DP, et al. Superconducting and magnetotransport properties of ZnNNi₃ microfibers and films. *Phys Rev* B. 2009;79:212508.
- Shim JH, Won SKK, Min BI. Electronic structures of antiperovskite superconductors MgXNi₃ (X=B, C, and N). *Phys Rev B*. 2001;64:180510.
- 21. Uehara M, Uehara A, Kozawa K, et al. Physica C. 2009;470:S688.
- Cao WH, Hea B, Liao CZ, et al. Preparation and properties of antiperovskite-type nitrides: InNNi₃ and InNCo₃. *J Solid State Chem*. 2009;182:3353.
- Chen WG, Wang F, Li SG, et al. Photoluminescence properties of Eu³⁺ and Bi³⁺ in YBO³ host under vacuum ultraviolet/ultraviolet excitation. *J Appl Phys*. 2009;105:123921.
- Shein IR, Bannikov VV, Ivanovskii AL. Elastic and electronic properties of the new perovskite-like superconductor ZnNNi₃ in comparison with MgCNi₃. *Phys Status Solidi B*. 2010;247:72.
- Okoye CMI. Structural, elastic and electronic properties of new antiperovskite-type superconductor from first-principles. *Physica B*. 2010;405:1562.
- 26. Cun ZH, Feng LX, Yi DJ, et al. First-principles study of mechanical stability and thermal properties of MNNi₃ (M = Zn, Mg, Al) under pressure. *Chin Phys B*. 2012;21:057102.
- Bannikov VV, Shein IR, Ivanovskii AL. Structural, elastic and electronic properties of new antiperovskite-like ternary nitrides AlNNi₃, GaNNi₃ and InNNi₃ as predicted from first principles. *Comput Mater Sci.* 2010;49:457.
- 28. Bannikov VV, Shein IR, Ivanovskii AL. Elastic properties of antiperovskite-type Ni-rich nitrides MNNi₃ (M=Zn, Cd, Mg, Al, Ga, In, Sn, Sb, Pd, Cu, Ag and Pt) as predicted from first-principles calculations. *Physica B*. 2010;405:4615.
- Helal MA, Islam AKMA. Elastic, electronic, and optical properties of hypothetical SnNN_{i3} and CuNNi₃ in comparison with superconducting ZnNNi₃. *Physica B*. 2011;406:4564.
- 30. Ali MA, Islam AKMA, Ali MS. J Sci Res. 2012;4(1):1-10.
- 31. Vanderbilt D. Soft self-consistent pseudopotentials in a generalized eigenvalue formalism. *Phys Rev B*. 1990;41:7892.
- 32. Kresse G, Hafner J. Norm-conserving and ultrasoft pseudopotentials for first-row and transition elements. *J Phys.: Condens. Matter.* 1994;6:8245.

- Hohenberg P, Khon W. Inhomogeneous Electron Gas. Phys Rev. 1964;136:B864.
- Khon W, Sham LJ. Self-Consistent Equations Including Exchange and Correlation Effects. *Phys Rev.* 1965;140:A1133.
- Perdew JP, Chevary JA, Vosko S, et al. Atoms, molecules, solids, and surfaces: Applications of the generalized gradient approximation for exchange and correlation. *Phys Rev B*. 1992;46:6671.
- Kresse G, Hafner J. Ab *initio* molecular dynamics for liquid metals. *Phys Rev B*. 1993;47:558.
- Kresse G, Hafner J. Ab *initio* molecular-dynamics simulation of the liquidmetal–amorphous-semiconductor transition in germanium. *Phys Rev B*. 1994;49:14251.
- 38. Kresse G, Furthmüller J. Efficiency of ab-*initio* total energy calculations for metals and semiconductors using a plane-wave basis set. *Comput Mat Sci.* 1996;6:15.
- Kresse G, Furthmüller J. Efficient iterative schemes for ab initio total-energy calculations using a plane-wave basis set. *Phys Rev B*. 1996;54:11169.
- Monkhorst HJ, Pack JD. Special points for Brillouin-zone integrations. *Phys Rev B*. 1976;13:5188.
- Marmier ZAD, Lethbridge RI, Walton CW, et al. ElAM: A computer program for the analysis and representation of anisotropic elastic properties. *Computer Physics Communications*. 2010;181:2102.
- Page L, Saxe P. Symmetry-general least-squares extraction of elastic data for strained materials from ab initio calculations of stress. *Phys Rev B*. 2002;65:04104.
- Mehl MJ, Osburn JE, Papaconstantopoulos DA, et al. Structural properties of ordered high-melting-temperature intermetallic alloys from firstprinciples total-energy calculations. *Phys Rev B*. 1990;41:10311.
- 44. Murnaghan FD. The Compressibility of Media under Extreme Pressures. *Proc Natl Acad Sci USA*. 1994;30:5390.
- 45. Xie YP, Wang ZY, Hou ZF. Scr Mater. 2013;68:495.
- Born M, Huang K. Dynamical Theory of Crystal Lattices. Clarendon Press; 1956.
- 47. Mayer B, Anton H, Bott E, et al. Ab-*initio* calculation of the elastic constants and thermal expansion coefficients of Laves phases. *Intermetallics*. 2003;11:23.
- Bannikov VV, Shein IR, Ivanovskii AL. Electronic structure, chemical bonding and elastic properties of the first thorium-containing nitride perovskite TaThN₃. Phys Status Solidi Rapid Reasecrh Letter. 2007;1(3):89.
- Rajeswarapalanichamiy R, Priyanga GS, Cinthia AJ, et al. Structural stability, electronic structure and mechanical properties of ZnN and CdN: A first principles study. *Computational Materials Science*. 2015;99:117.
- 50. Nye JF. Physical Properties of Crystals. Oxford: Clerandon; 1985.
- Xu JH, Oguchi T, Freeman AJ. Crystal structure, phase stability, and magnetism in Ni3V. *Phys Rev B*. 1987;35:6940.
- 52. Xu JH, Freeman AJ. Band filling and structural stability of cubic trialuminides: YAl₃, ZrAl₃, and NbAl₃. *Phys Rev B*. 1989;40:11927.
- Toga A, Oba F, Tanaka I. First-principles calculations of the ferroelastic transition between rutile-type and CaCl₂-type SiO₂ at high pressures. *Phys Rev B*. 2008;78:134106.